Thermal transport in aluminum oxide irradiated with swift heavy ions

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Outline
- MOTIVATION
- METHOD
- SIMULATION
- RESULT
- SUMMARY

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**MOTIVATION**

**EFFECT OF SWIFT HEAVY IONS**

Ion energy loss:
- Electronic $S_e$
- Nuclear $S_n$

$\text{Sn/Se} \leq 0.01$

$E > 0.5 \text{ MeV/nucl}$,

$M > 4m_p$

- Bi ion in Al$_2$O$_3$

**THERMAL CONDUCTIVITY DEGRADATION**

- The resistance at irradiation with $n$, alpha and fission fragments;
- Good thermal properties
- The aluminum oxide (Al$_2$O$_3$) is a promising material for various nuclear applications, it is considered as a candidate material for inert matrix fuel host in the reactor core;
- Study thermal conductivity degradation after irradiation with fission fragments is an important issue of materials for future high temperature nuclear reactors.

**Simulate an influence of fission fragments and cosmic rays on material properties:**

**Nanostructuring of solids**
**METHOD**

**Molecular Dynamics**

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**Initialization:**
- set for each particle \( i \)
- position \( \mathbf{r}_i(t_0) \)
- velocity \( \mathbf{v}_i(t_0) \)

**Time Step:**
- update for each particle \( i \)
  - \( \mathbf{r}_i(t) \rightarrow \mathbf{r}_i(t + \Delta t) \)
  - \( \mathbf{v}_i(t) \rightarrow \mathbf{v}_i(t + \Delta t) \)

**Loop Over Time Steps**

**Initial conditions for simulation with MD**

Crystal structure (Unit cell)

Interatomic potential

\[ \rho = 3.99 \ \text{g/cm}^3 \]

**MD based on solving Newton equations:**

\[ \mathbf{F} = -\nabla V \]

\[ m_i \frac{d^2 \mathbf{r}_i(t)}{dt^2} = F_i(r), \quad i = 1,2,\ldots,n. \]

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Al\(_2\)O\(_3\) has a trigonal syngony with the space group R3c. The crystal lattice consists of Al\(^{3+}\) and O\(^{2-}\). Oxygen ions form a distorted hexagonal dense packing, and aluminum cations occupy 2/3 of octahedral voids.

Three-body Vashishta-type potential[1]

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1) Track formation = TREKIS + MD

TREKIS: Monte Carlo simulation

- Swift Heavy ion impact
- Electronic kinetics
- Atomic dynamics

Molecular Dynamics Simulation

- Macroscopic relaxation

Further this structure was used as input data for simulation thermal conductivity


https://www.ovito.org/
2) Direct method for calculation thermal conductivity (Fourier law)

I. Creation of non-equilibrium system

II. Transition of the system to a stationary state; Computation of thermal conductivity.

III. We used the simulation cells varying from 29 to 330 nm (22-250 unit cells) in length.

This problem is solved with the well-developed method of thermal conductivity calculations in several simulation cells of different sizes with subsequent extrapolation of its value to an infinitely large system, \( k_\infty \), according to the Mathiessen’s rule.
RESULT

1) Change of thermal conductivity after SHI irradiation

(a) MD simulated track of 167 MeV Xe ion in alumina. The squares show cut cells for different fluences. Blue dots with numbers shows the positions of the ion trajectories in order of its simulations. (b) Selected central area with simulated several subsequent ion passages. (c) Plot of inverse thermal conductivity vs inverse cell length for different fluences.

\[
Fluence = \frac{N}{S}
\]

(a) Fluenec dependence of the calculated thermal conductivity coefficient of alumina. Experimental data and their fit with the phonon-mediated thermal transport model. (b) Thickness of amorphous layer for Xe ions of different energies from Kabir et al., Abdullaev et al., Okubo et al.. Inset shows the scheme of layered structure of Al2O3 irradiated with high fluences.
(a) MD image of 167 MeV Xe ion track in Al2O3. Red square shows the cut of the track core of 2 × 2 nm² in size; (b) pristine matrix and (c) track core inserted into pristine matrix in order to eliminate effects of track halo; (d) MD image of track core in pristine matrix relaxed at 300 K with NPT ensemble; (e) Finite-size scaling of calculated thermal conductivity (with linear fits) of MD cells shown in (a) and (d).
(a) MD image of three subsequent 167 MeV Xe tracks in Al2O3. (b) pristine matrix with inserted three track cores; (c) MD image of three track cores in pristine matrix relaxed at 300 K with NPT ensemble. (d) Finite-size scaling of calculated thermal conductivity (with linear fits) of MD cells shown in (a) and (c).
SUMMARY

• For the first time thermal transport was simulated in SHI irradiated solid using a direct method, which demonstrated good applicability in this case. Degradation of the thermal conductivity of single-crystalline Al2O3 with fluence demonstrates good agreement with the experimental TDTR data.

• The applied method allowed to separate an effect of discontinuous crystalline cylindrical tracks from that of an amorphous surface layer on the thermal conductivity of irradiated alumina. The results demonstrate that a very narrow highly damaged track core, created by SHI, most strongly affects the heat transport in such targets.

• The obtained data can be used for prediction of long-term radiation stability and thermal properties of materials exposed with fission fragments. They also can be used as the basis for the design of thermo-electronic devises with tailored properties and for advanced technologies of their productions.
Acknowledgement

This work has been carried out using next computing resources:
1) Federal collective usage center Complex for Simulation and Data Processing for Mega-science Facilities at NRC “Kurchatov Institute”
2) GSI Helmholtzzentrum (Darmstadt, Germany)
3) HybriLIT heterogeneous computing platform (LIT, JINR)
### Appendix 1: Interatomic potential

\[
V = \sum_{i<j} V^{(2)}_{ij}(r_{ij}) + \sum_{i,j<k} V^{(3)}_{ijk}(r_{ij}, r_{ik}).
\]

\[
V^{(2)}_{ij}(r) = \frac{H_{ij}}{r^{\eta}} + \frac{Z_i Z_j}{r} e^{-r/\lambda} - \frac{D_{ij}}{r^4} e^{-r/\xi} - \frac{W_{ij}}{r^6}.
\]

**Spatial factor**

**Angular factor**

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**Steric-size effects interactions**

**Coulomb interactions**

**Charge-induced dipole**

**Van der Waals interactions**

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**TABLE II. Parameters for two- and three-body parts of the interaction potential used in the MD simulation of structural and dynamical properties of amorphous and liquid Al\(_2\)O\(_3\).**

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<th>Al</th>
<th>O</th>
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<tr>
<td>(Z_i (e))</td>
<td>1.5237</td>
<td>-1.0158</td>
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<tr>
<td>(\lambda = 5.0 \text{ Å})</td>
<td>3.75 Å</td>
<td>6.0 Å</td>
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<tr>
<td>(\xi = 6.0 \text{ Å})</td>
<td>1.602 \times 10^{-19} C</td>
<td></td>
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<table>
<thead>
<tr>
<th></th>
<th>Al–Al</th>
<th>Al–O</th>
<th>O–O</th>
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<tbody>
<tr>
<td>(\eta_{ij})</td>
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<td>9</td>
<td>7</td>
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<tr>
<td>(H_{ij} (eV \text{ Å}^3))</td>
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<tr>
<td>(D_{ij} (eV \text{ Å}^3))</td>
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<td>(W_{ij} (eV \text{Å}^3))</td>
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<table>
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<th>(\tilde{\theta}_{ijk} (\text{deg}))</th>
<th>(C_{ijk})</th>
<th>(\gamma (\text{Å}))</th>
<th>(r_0 (\text{Å}))</th>
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<td>O–Al–O</td>
<td>12.4844</td>
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**From paper**

Interaction potentials for alumina and molecular dynamics simulations of amorphous and liquid alumina

**Priya Vashishta, Rajiv K. Kalia, Aiichiro Nakano, and José Pedro Rino**

**JOURNAL OF APPLIED PHYSICS** 103, 083504v (2008)

[https://doi.org/10.1063/1.2901171](https://doi.org/10.1063/1.2901171)
Appendix 2: TREKIS and MD

**Time-Resolved Electron Kinetics in SHI Irradiated Solids**

- (a) passage of a swift heavy ion ionizing the target and generating primary electrons and holes;
- (b) interaction of δ-electrons with lattice and target electrons and further secondary electronic cascades;
- (c) Auger decays of deep shell holes also producing secondary electrons;
- (d) Radiative decays of core holes, following photon transport and photoabsorption exciting new electrons and holes;
- (e) Valence, holes redistribution and their interaction with electrons and lattice.

The dynamic structure factor - complex dielectric function formalism is applied to build up cross sections describing charged particles interactions within TREKIS.

\[
\frac{d^2\sigma}{d(h\omega)d(hq)} = \frac{2[Z_e(v, q)e]^2}{\pi h^2 v^2} \frac{1}{hq} \operatorname{Im} \left[ \frac{-1}{\varepsilon(\omega, q)} \right].
\]

**From paper**

Time-resolved electron kinetics in swift heavy ion irradiated solids

N A Medvedev, R A Rymzhanov and A E Volkov


http://dx.doi.org/10.1088/0022-3727/48/35/355303
Appendix 3: Direct method TC

\[
\frac{1}{l_{\text{eff}}} = \frac{1}{l_{\infty}} + \frac{4}{L_z}.
\]

\[
\kappa = \frac{1}{3} \text{ cv}1,
\]

\[
\frac{1}{\kappa} = \frac{a^3}{4k_B v} \left( \frac{1}{l_{\infty}} + \frac{4}{L_z} \right).
\]

\[
c = \frac{3}{2} k_B n,
\]

From paper

Comparison of atomic-level simulation methods for computing thermal conductivity

Patrick K. Schelling, Simon R. Phillpot, and Pawel Keblinski

PHYSICAL REVIEW B, VOLUME 65, 144306

https://doi.org/10.1103/PhysRevB.65.144306
Appendix 4: Experimental details of TC

Picosecond time domain thermoreflectance (TDTR)

To perform near surface nanoscale thermal transport measurement on irradiated sapphire samples, have implemented the TDTR setup. The Ti:Al2O3 mode-locked femtosecond laser (Tsunami, Spectra Physics) at 782 nm wavelength, 80 MHz repetition rate and 80 fs pulse duration, was used as pump and probe beams.

Recorded picosecond TDTR signal of $-V_{in}/V_{out}$ as a function of delay time between pump and probe beams was fitted with a thermal diffusion model to extract unknown thermal conductivity. The thermal analysis was based on the iterative algorithm applied to a multilayer geometry with thermal conductivity, volumetric specific heat capacity, and thickness defined for each layer.

From paper

Thermal transport across nanoscale damage profile in sapphire irradiated by swift heavy ions

A. Abdullaev, V. S. Chauhan, B. Muminov, J. O’Connell, V. A. Skuratov, M. Khafizov and Z. N. Utegulov


https://doi.org/10.1063/1.5126413